

Orbital Engineering in Nickelate Heterostructures Driven by Anisotropic Oxygen Hybridization rather than Orbital Energy Levels

G. Fabbri,^{1,*} D. Meyers,¹ J. Okamoto,² J. Pellicciari,³ A. S. Disa,⁴ Y. Huang,³ Z.-Y. Chen,² W. B. Wu,² C. T. Chen,² S. Ismail-Beigi,^{4,5} C. H. Ahn,^{4,5} F. J. Walker,⁴ D. J. Huang,^{2,6} T. Schmitt,³ and M. P. M. Dean^{1,†}

¹*Department of Condensed Matter Physics and Materials Science, Brookhaven National Laboratory, Upton, New York 11973, USA*

²*National Synchrotron Radiation Research Center, Hsinchu 30076, Taiwan*

³*Research Department “Synchrotron Radiation and Nanotechnology”, Paul Scherrer Institut, CH-5232 Villigen PSI, Switzerland*

⁴*Department of Applied Physics, Yale University, New Haven, Connecticut 06520, USA*

⁵*Department of Mechanical Engineering and Materials Science, Yale University, New Haven, Connecticut 06520, USA*

⁶*Department of Physics, National Tsing Hua University, Hsinchu 30013, Taiwan*

(Received 17 March 2016; revised manuscript received 27 May 2016; published 30 September 2016)

Resonant inelastic x-ray scattering is used to investigate the electronic origin of orbital polarization in nickelate heterostructures taking $\text{LaTiO}_3 - \text{LaNiO}_3 - 3 \times (\text{LaAlO}_3)$, a system with exceptionally large polarization, as a model system. We find that heterostructuring generates only minor changes in the Ni $3d$ orbital energy levels, contradicting the often-invoked picture in which changes in orbital energy levels generate orbital polarization. Instead, O K -edge x-ray absorption spectroscopy demonstrates that orbital polarization is caused by an anisotropic reconstruction of the oxygen ligand hole states. This provides an explanation for the limited success of theoretical predictions based on tuning orbital energy levels and implies that future theories should focus on anisotropic hybridization as the most effective means to drive large changes in electronic structure and realize novel emergent phenomena.

DOI: 10.1103/PhysRevLett.117.147401

The electronic structure of transition metal oxides (TMOs) is dominated by the active $3d$ TM orbitals and how these hybridize with the neighboring oxygen $2p$ ligand orbitals. Building heterostructures from one-unit-cell-thick layers of different TMOs offers the opportunity of tuning the TM $3d$ states configuration and potentially realizing emergent phenomena with new or improved properties [1–6]. LaNiO_3 based heterostructures are a prototypical example of such an endeavor [6–20], motivated by several predictions of novel superconducting, magnetic, and topological states [21–24]. Bulk LaNiO_3 has *nominally* Ni^{3+} ions with filled t_{2g} states and one e_g electron. Efforts to realize the predicted novel states are crucially dependent on inducing orbital polarization in the Ni e_g level [25], i.e., breaking its degeneracy and pushing the system towards a half filled $x^2 - y^2$ (or $3z^2 - r^2$) configuration [Fig. 1(b)]. Initial efforts to realize strong orbital polarization in heterostructures reported up to $\sim 20\%$ change in orbital occupancy compared to the bulk [7–12,14,19], corresponding to ~ 0.3 eV splitting of the e_g states as inferred from cluster calculations of x-ray absorption spectra [14]; in fact, first principle calculations indicate that $\sim 10\%$ strain is needed to drive $\sim 40\%$ change in orbital polarization (see Supplemental Material of Ref. [17]). Novel trilayer $\text{LaTiO}_3 - \text{LaNiO}_3 - 3 \times (\text{LaAlO}_3)$ (LTNAO) superlattices were recently produced obtaining $\sim 50\%$ change in orbital polarization via charge transfer, polar charge, and electronic confinement effects [15–18]. To date, changes in orbital polarization

have been conceptualized as a consequence of tuning the $3d$ orbital energy level through crystal field engineering. However, this approach has led to a dramatic mismatch between theoretical predictions and experimental results [14,21–23,30], raising questions about the precise electronic character, and the driving mechanism behind, orbital polarization in these materials.

We apply resonant inelastic x-ray scattering (RIXS) [31,32] to LaNiO_3 and LTNAO superlattices obtaining a far more detailed spectral fingerprint of the Ni electronic configuration than is possible with standard x-ray absorption spectroscopy (XAS) [7–9,12,14,17,19,33]. Charge transfer is found to drive LaNiO_3 out of its initial itinerant state in the bulk into a more localized state in LTNAO superlattices that display Ni $d^8 \underline{L}$ electronic configuration (where \underline{L} denotes partial oxygen ligand hole character). A crystal field splitting $\Delta e_g = 0.20(5)$ eV is found in LTNAO superlattices, far smaller than previously thought and inconsistent with the often invoked picture of orbital polarization driven by crystal field splitting [8,14,15,17,21,22]. X-ray absorption spectroscopy at the O K edge demonstrates that orbital polarization is instead driven by anisotropic hybridization with the oxygen ligands. This explains the current lack of success in realizing novel emergent phenomena in nickelate heterostructures based on changing orbital energy levels [21–24]. Our work indicates that manipulating the anisotropy of the Ni hybridization with oxygen will be the most effective route to drive large changes in the electronic structure of

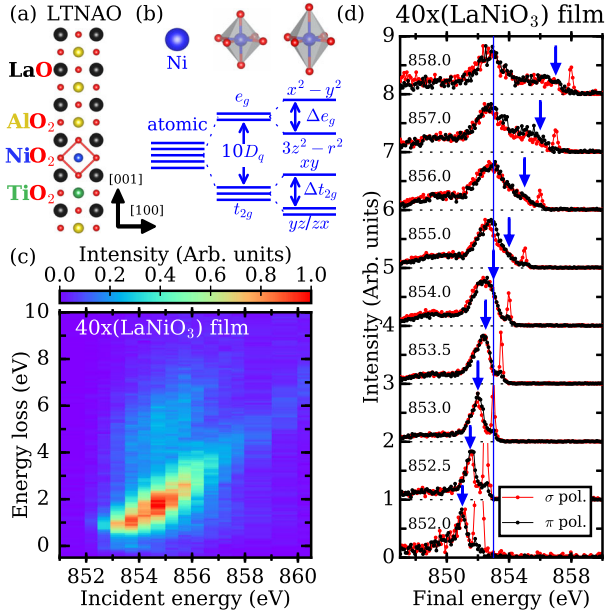


FIG. 1. (a) The basic structural unit studied here composed of LaTiO_3 , LaNiO_3 , and LaAlO_3 layers in a 1:1:3 ratio. (b) Energy level diagram for the Ni electronic state. (c) Ni L_3 -edge RIXS map of a LaNiO_3 film measured with π -polarized incident x rays. The strongest inelastic spectral feature is the diagonal constant final energy emission line characteristic of a metallic system. (d) Normalized RIXS spectra as a function of final energy, for different incident energies, and both polarizations as labeled. A vertical blue line at 853 eV marks the fluorescence feature. Blue arrows denote 1.1 eV energy loss. The sharp feature observed at larger final energies for σ is the elastic line. For incident energies, 852–853.5 eV and 857–859 eV a 1.1 eV constant energy loss feature is visible.

nickelates and therefore provides the best chance of realizing novel emergent states.

Samples were synthesized on LaAlO_3 [001] substrates using oxygen plasma-assisted molecular beam epitaxy (MBE) as described in detail in Ref. [17]. The LTNAO sample is depicted in Fig. 1(a) and was prepared with a layering sequence $[\text{LaTiO}_3 - \text{LaNiO}_3 - 3 \times (\text{LaAlO}_3)] \times 3$. The LaNiO_3 sample was a 40 formula unit thick reference. The LaAlO_3 substrate introduces a $\sim 1.1\%$ compressive strain in LaNiO_3 . X-ray characterization shows excellent structural quality with roughnesses of approximately 5 \AA [26]. We determined the dichroism in the XAS signal of LTNAO by integrating our RIXS data along the energy loss axis and found $\sim 50\%$ increase in the in-plane character of the orbital polarization [26] consistent with previous work [17].

During the RIXS experiments, a $(H, 0, L)$ horizontal scattering plane was employed. X rays were incident at 15° with respect to the sample surface polarized either parallel (π) or perpendicular (σ) to the scattering plane and x rays scattered around 90° were energy analyzed by the spectrometer [26]. The LTNAO sample was measured with the SAXES spectrometer [34] located at the ADDRESS

beam line [35] at the Swiss Light Source of the Paul Scherrer Institut and the LaNiO_3 sample was measured with the AGS-AGM setup [36] at BL05A1—the inelastic scattering beam line at the National Synchrotron Radiation Research Center, Taiwan. The combined energy-loss resolution for both experiments was $\sim 210 \text{ meV}$. All data were collected at $\sim 13 \text{ K}$ to reduce thermal diffuse scattering.

When grown in the bulk, LaNiO_3 , the active constituent of the heterostructures studied here, is a correlated metal with approximately cubic symmetry [37]. The nominal Ni^{3+} ground state is usually interpreted as a mixture between d^7 and $d^8\bar{L}$ electronic configurations [12,19,38–43]. Figure 1(c) plots a map of the RIXS intensity of the LaNiO_3 film. The main spectral feature appears as a diagonal, constant final energy emission line, that is visible for incident energies above the Ni L_3 resonance around 853 eV, as is typically observed for systems with mostly metallic character [44–46]. Figure 1(d) plots individual spectra as a function of final energy, normalized to peak intensity, which reveals a weak Raman-like constant energy loss feature at $\sim 1.1 \text{ eV}$, suggesting a partially localized component to the electronic character. Recent RIXS and theoretical work place LaNiO_3 in the negative charge transfer regime [43,47–53], with RIXS Raman features coming from $3d^8\bar{L}$ states. The polarization dependence of the $\sim 1.1 \text{ eV}$ feature is directly related to the e_g splitting [26]. Therefore, the weak polarization dependence shown in Fig. 1(d) places an upper limit on the strain-induced e_g splitting $\Delta e_g \lesssim 0.2 \text{ eV}$ [26] in agreement with the literature [7–12,14].

Figures 2(a) and 2(b) plot the dramatic change in the Ni L_3 edge RIXS maps once LaNiO_3 is confined within a LTNAO heterostructure. Rather than constant final energy fluorescence, the spectra now feature Raman-like constant energy loss excitations that resonate differently as a function of incident energy. In this regard, the Ni electronic state looks much more similar to correlated insulators with a Ni d^8 configuration such as NiO or NiCl_2 [54–56]. Indeed, LTNAO has been observed to be far more electrically insulating than LaNiO_3 [17]. By measuring both the incident and scattered x-ray photon energies, RIXS obtains many more distinct spectral features than XAS, particularly because the linewidth of the RIXS final states are not limited by the core hole lifetime. We analyzed our data using multiplet calculations in the Cowan-Butler-Thole approach [57–59], which computes the L -edge RIXS signal for an ion within a given crystal field. These methods are widely used to model resonant x-ray spectra of TMOs [58] and are explained in more detail in the Supplemental Material [26].

We found that the LTNAO RIXS can be modeled in terms of dd excitations from a Ni d^8 atom in a tetragonal crystal field as plotted in Fig. 2 with linecuts shown in Fig. 3. The parameters describing the LTNAO electronic configuration are detailed in Table I, and Fig. 1(b)

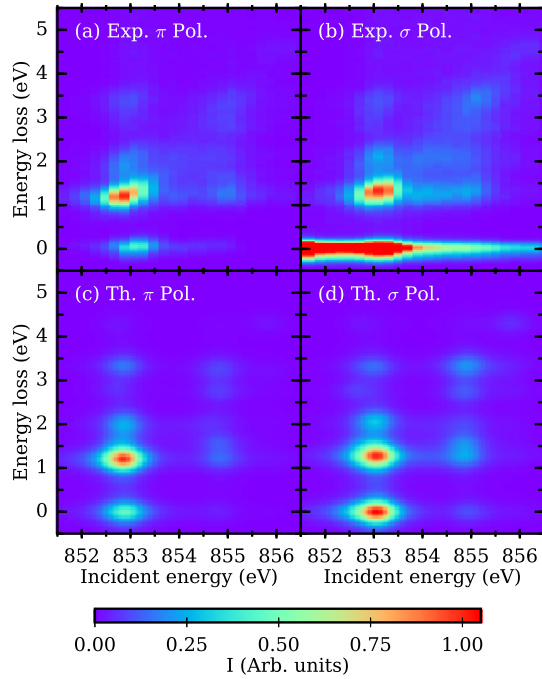


FIG. 2. Color maps of the RIXS intensity of the LTNAO heterostructure. The spectra are dominated by Raman-like constant energy loss excitations that resonate differently as a function of incident energy, providing a detailed spectral fingerprint of the Ni electronic structure in the heterostructure. (a) and (b) plot the measured intensity with π and σ polarized incident x rays. The strong elastic peak at low incident energies for σ polarization is due to the La M_4 edge. (c) and (d) plot our corresponding multiplet calculations for the two different polarizations.

illustrates the parameters $10D_q$, Δt_{2g} , and Δe_g in terms of the orbital energy levels. Details on the estimated errors are described in the Supplemental Material [26]. The atomic model used here assigns only d states to the valence band; hence, the extracted energies correspond to those of effective hybridized orbitals in the electronic structure. We find excellent agreement between experiment and theory for the peak energies, as good, or better, than multiplet calculations for much simpler bulk compounds [54–56]. A slightly poorer agreement is found regarding the intensity of the excitations, which is likely due to small intensity renormalizations driven by the presence of ligand holes [55]. Nevertheless, RIXS shows that heterostructuring has driven the system closer to a Ni d^8 electronic configuration than a Ni d^7 configuration (d^7 calculations completely fail to reproduce our measurements [26]).

The predominant Ni d^8 electronic configuration is likely to arise primarily from charge transfer effects, as unlike some other perovskites, the $3d$ levels in LaTiO₃ lie near to the Fermi level of LaNiO₃ heterostructures [60,61]. Ti L -edge XAS showed that the Ti charge state changes from 3+ to close to 4+ upon heterostructuring, indicating that a large fraction of the one electron per Ti has been transferred

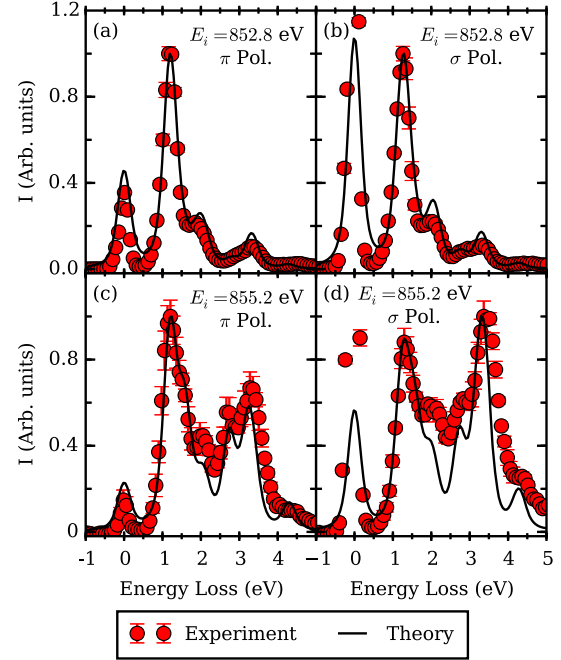


FIG. 3. Comparison of the experimental and theoretical RIXS spectra of the LTNAO heterostructure at fixed incident energy. Peak energies are reproduced within about 0.1 eV and intensities are reproduced within $\sim 20\%$. (a) and (b) plot spectra at an incident energy of $E_i = 852.8$ eV for π and σ polarized incident x rays, respectively. (c) and (d) plot the same quantities for $E_i = 855.2$ eV. Note that for σ incident x rays the elastic line intensity is a combination of RIXS intensity and diffuse scattering not included in the model, so full agreement is not necessarily expected.

to other layers in the heterostructure [17] driving Ni towards a d^8 state.

Orbital polarization in nickelate heterostructures is usually interpreted to be due to enhanced Δe_g driven by tetragonal distortions [8,14,15,17,21,22]. The ratio between out-of- and in-plane Ni-O bond length is as large as 1.16 in LTNAO [17], an anisotropy comparable to that of La₂NiO₄, for which $\Delta e_g \sim 0.7$ eV [62]. Indeed, $\Delta e_g \sim 0.8$ eV was suggested previously by projecting the DFT band structure onto effective hybridized $3z^2 - r^2$ and $x^2 - y^2$ Wannier orbitals [17]. However, the effective Ni

TABLE I. The parameters describing the electronic configuration of Ni in the LTNAO heterostructure. $10D_q$, Δt_{2g} , and Δe_g describe the crystal field energy levels [see Fig. 1(b)], and F and G control the values of the Slater-Condon parameters [58]. F_{dd} and F_{pd} are the percentage reductions in the Coulomb repulsion between two d electrons and p - d electrons, respectively. G_{pd} is the percentage reduction in the Coulomb exchange. Errors were calculated as described in the Supplemental Material [26].

	$10D_q$	Δt_{2g}	Δe_g	F_{dd}	F_{pd}	G_{pd}
Values	1.28(3) eV	0.05(5) eV	0.20(5) eV	65%	95%	70%

crystal field levels measured by RIXS in LTNAO yields $\Delta e_g = 0.20(5)$ eV, at most ~ 0.1 eV larger than what is induced by strain alone. The large reconstruction of oxygen ligand holes (discussed below) may explain such unexpectedly weak structural dependence of Δe_g , but further work is needed to address this result. Nevertheless, our work demonstrates the ability of Ni L_3 -edge RIXS in obtaining a precise description of the crystal field energies acting upon $3d$ orbitals in heterostructures. Coupled with ongoing efforts to improve theoretical calculations [31,59,63–67], this emphasizes the as yet underutilized potential of this technique to understand the properties of transition metal oxide heterostructures.

In the face of LTNAO's small tetragonal crystal field splitting, the origin of Ni's orbital polarization needs to be revisited. Although Ni L -edge RIXS provides an accurate picture of orbital energy levels, it is much less sensitive to the degree of oxygen ligand hole character. In particular, the multiplet calculations used here yield a Ni d^8 high spin state, with formally one electron in the $3z^2 - r^2$ and $x^2 - y^2$ orbitals, and cannot explain the observed orbital polarization. The multiplet calculations, however, do not include the possibility of oxygen ligand holes, which can modify the effective orbital occupation. We therefore examined the oxygen ligand hole states by O K -edge XAS plotted in Fig. 4. Every layer of the sample contains oxygen atoms that contribute to the main absorption edge and to features above the edge. On the other hand, preedge peaks that provide information about hybridization can be distinguished by comparing their energies to the literature [19,68–70]. The prepeak at ≈ 528.5 eV is assigned to Ni $3d^8\bar{L}$ states in the LaNiO_3 layers, while the ≈ 531 eV features come from Ni $3d^9\bar{L}$ states and Ti states in the LaTiO_3 layers [19]. Thus, Fig. 4 indicates that the Ni d^8 ground state features partial Ni $d^8\bar{L}$ character, likely coming from an incomplete $\text{Ti} \rightarrow \text{Ni}$ charge transfer.

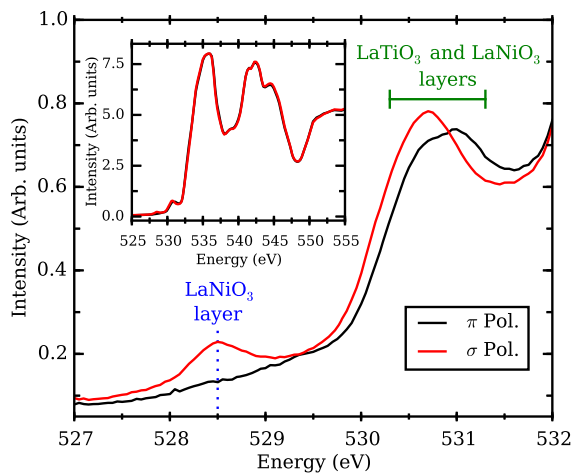


FIG. 4. LTNAO O K -edge x-ray absorption spectra measured in total fluorescence yield mode. The inset shows the full energy window, the preedge region of which is plotted in the main panel.

Ligand holes appear to only weakly alter Ni L_3 -edge RIXS, consistent with Anderson impurity model calculations for NiO [55].

The intensity of the 528.5 eV LaNiO_3 prepeak is proportional to the number of ligand holes [68–72]. Remarkably, a very large linear dichroism is observed in this prepeak (Fig. 4), the π channel is at least $20\times$ weaker than σ (in the former any signal is within the measurement error). This large anisotropy translates into more holes in the $3d_{x^2-y^2}p_\sigma$ hybridized state than in the $3d_{3z^2-r^2}p_z$ state, where p_σ and p_z denote in-plane and out-of-plane oxygen states, establishing that the large orbital polarization measured at the Ni edge originates from a uniquely anisotropic $3d^8\bar{L}$ state. The ligand hole anisotropy in LTNAO (as measured by XAS) is much closer to that observed in cuprates (~ 10 – $100\times$) [71,72], than in nickelates (~ 1 – $2\times$) [68,69]. Therefore, heterostructuring successfully induced a dominating $3d_{x^2-y^2}p_\sigma$ character to LaNiO_3 Fermi surface. Such character is among the defining properties of cuprate superconductors, thus placing LTNAO as the closest $S = 1$ analogue to cuprates known to date. Probing the fermiology of LTNAO, particularly upon hole doping, remains as an outstanding opportunity for future investigations.

In conclusion, we demonstrate that the large orbital polarization observed in LTNAO is driven by an anisotropic hybridization of the $3d - 2p$ orbitals. This result largely disagrees with the common attempt to manipulate the properties of nickelate heterostructures by tuning the crystal field symmetry, an approach that has led to exciting predictions, but which have failed to materialize [14,21–23,30]. On the other hand, our results are congruent with the growing realization that LaNiO_3 resides in the negative charge transfer regime, dominated by the $3d^8\bar{L}$ configuration [20,43,47–51]. The $3d^8$ high spin triplet is very stable. Because of the large Hund's and on site Coulomb energies, a very large tetragonal splitting, of the order of $10D_q$ [1.28(3) eV in LTNAO] is required to destabilize this ground state. A similar situation is realized in $\text{La}_2\text{NiO}_{4+\delta}$ ($\delta = 0, 0.05, 0.12$) which has negligible ($< 5\%$) orbital polarization despite its very large $\Delta e_g \sim 0.7$ eV [62]. Ni L -edge x-ray linear dichroism measurements of heavily doped $\text{La}_{2-x}\text{Sr}_x\text{NiO}_4$ quantifying possible orbital polarization in this system and comparing it to LTNAO would also be a highly interesting reference.

Inducing substantial orbital polarization in the Ni $3d$ state appears to be only achievable through anisotropic Ni-O hybridization. This implies that the origin of the orbital polarization in various nickelate heterostructures needs to be reexamined. Furthermore, a number of emergent phenomena occur in systems located in the small or negative charge transfer regime of the Zaanen-Sawatzky-Allen phase diagram [73–75], in which ligand holes are the dominant state at the Fermi surface. Examples of such systems include CrO_2 [76], BaFeO_3 [77], SrCoO_3 [78], and cuprates [74]. For instance, in cuprates, the ligand hole

couples to the $3d$ hole to form a Zhang-Rice singlet [79], a state that is believed to be crucial for high- T_c superconductivity. Therefore, the ability to control the TM $3d$ —O $2p$ hybridization through charge transfer, polar charge, and electronic confinement effects, in tandem with first principles calculations that treat strong correlations and oxygen hybridization accurately such as dynamical mean field theory [53], represents a highly promising route towards achieving the as yet largely unrealized aim of generating novel emergent states in transition metal oxide heterostructures.

This material is based upon work supported by the U.S. Department of Energy, Office of Basic Energy Sciences, Early Career Award Program under Award No. 1047478. We thank Frank de Groot, Valentina Bisogni, Liviu Hozoi, Jeroen van den Brink, Eli Stavitski, Jian Liu, and John Tranquada for helpful discussions. J. P. and T. S. acknowledge financial support through the Dysenos AG by Kabelwerke Brugg AG Holding, Fachhochschule Nordwestschweiz, and the Paul Scherrer Institut. Work at Yale was supported by NSF under MRSEC DMR-1119826 (CRISP) and by AFOSR. Experiments were performed at the ADRESS beam line of the Swiss Light Source at the Paul Scherrer Institut, Switzerland and at BL05A1—Inelastic Scattering at National Synchrotron Radiation Research Center, Taiwan.

*gfabbris@bnl.gov

†mdean@bnl.gov

- [1] D. G. Schlom, L.-Q. Chen, X. Pan, A. Schmehl, and M. A. Zurbuchen, *J. Am. Ceram. Soc.* **91**, 2429 (2008).
- [2] P. Zubko, S. Gariglio, M. Gabay, P. Ghosez, and J.-M. Triscone, *Annu. Rev. Condens. Matter Phys.* **2**, 141 (2011).
- [3] H. Hwang, Y. Iwasa, M. Kawasaki, B. Keimer, N. Nagaosa, and Y. Tokura, *Nat. Mater.* **11**, 103 (2012).
- [4] A. Bhattacharya and S. J. May, *Annu. Rev. Mater. Res.* **44**, 65 (2014).
- [5] J. Chakhalian, J. W. Freeland, A. J. Millis, C. Panagopoulos, and J. M. Rondinelli, *Rev. Mod. Phys.* **86**, 1189 (2014).
- [6] S. Middey, J. Chakhalian, P. Mahadevan, J. Freeland, A. Millis, and D. Sarma, *Annu. Rev. Mater. Res.* **46**, 305 (2016).
- [7] E. Benckiser, M. W. Haverkort, S. Brück, E. Goering, S. Macke, A. Frañó, X. Yang, O. K. Andersen, G. Cristiani, H.-U. Habermeier, A. V. Boris, I. Zegkinoglou, P. Wochner, H.-J. Kim, V. Hinkov, and B. Keimer, *Nat. Mater.* **10**, 189 (2011).
- [8] J. Chakhalian, J. M. Rondinelli, J. Liu, B. A. Gray, M. Kareev, E. J. Moon, N. Prasai, J. L. Cohn, M. Varela, I. C. Tung, M. J. Bedzyk, S. G. Altendorf, F. Strigari, B. Dabrowski, L. H. Tjeng, P. J. Ryan, and J. W. Freeland, *Phys. Rev. Lett.* **107**, 116805 (2011).
- [9] J. W. Freeland, J. Liu, M. Kareev, B. Gray, J. W. Kim, P. Ryan, R. Pentcheva, and J. Chakhalian, *Europhys. Lett.* **96**, 57004 (2011).
- [10] M. J. Han, C. A. Marianetti, and A. J. Millis, *Phys. Rev. B* **82**, 134408 (2010).
- [11] M. J. Han, X. Wang, C. A. Marianetti, and A. J. Millis, *Phys. Rev. Lett.* **107**, 206804 (2011).
- [12] J. Liu, S. Okamoto, M. van Veenendaal, M. Kareev, B. Gray, P. Ryan, J. W. Freeland, and J. Chakhalian, *Phys. Rev. B* **83**, 161102 (2011).
- [13] M. Gibert, P. Zubko, R. Scherwitzl, J. Íñiguez, and J.-M. Triscone, *Nat. Mater.* **11**, 195 (2012).
- [14] M. Wu, E. Benckiser, M. W. Haverkort, A. Frano, Y. Lu, U. Nwankwo, S. Brück, P. Audehm, E. Goering, S. Macke, V. Hinkov, P. Wochner, G. Christiani, S. Heinze, G. Logvenov, H.-U. Habermeier, and B. Keimer, *Phys. Rev. B* **88**, 125124 (2013).
- [15] H. Chen, D. P. Kumah, A. S. Disa, F. J. Walker, C. H. Ahn, and S. Ismail-Beigi, *Phys. Rev. Lett.* **110**, 186402 (2013).
- [16] D. P. Kumah, A. S. Disa, J. H. Ngai, H. Chen, A. Malashevich, J. W. Reiner, S. Ismail-Beigi, F. J. Walker, and C. H. Ahn, *Adv. Mater.* **26**, 1935 (2014).
- [17] A. S. Disa, D. P. Kumah, A. Malashevich, H. Chen, D. A. Arena, E. D. Specht, S. Ismail-Beigi, F. J. Walker, and C. H. Ahn, *Phys. Rev. Lett.* **114**, 026801 (2015).
- [18] A. S. Disa, F. Walker, S. Ismail-Beigi, and C. H. Ahn, *APL Mater.* **3**, 062303 (2015).
- [19] Y. Cao, X. Liu, M. Kareev, D. Choudhury, S. Middey, D. Meyers, J. W. Kim, P. J. Ryan, J. W. Freeland, and J. Chakhalian, *Nat. Commun.* **7**, 10418 (2016).
- [20] M. N. Grisolia, J. Varignon, G. Sanchez-Santolino, A. Arora, S. Valencia, M. Varela, R. Abrudan, E. Weschke, E. Schierle, J. E. Rault, J.-P. Rueff, A. Barthélémy, J. Santamaria, and M. Bibes, *Nat. Phys.* **12**, 484 (2016).
- [21] J. Chaloupka and G. Khaliullin, *Phys. Rev. Lett.* **100**, 016404 (2008).
- [22] P. Hansmann, X. Yang, A. Toschi, G. Khaliullin, O. K. Andersen, and K. Held, *Phys. Rev. Lett.* **103**, 016401 (2009).
- [23] A. Rüegg and G. A. Fiete, *Phys. Rev. B* **84**, 201103 (2011).
- [24] K.-Y. Yang, W. Zhu, D. Xiao, S. Okamoto, Z. Wang, and Y. Ran, *Phys. Rev. B* **84**, 201104 (2011).
- [25] For the purposes of this Letter, we define orbital polarization in terms of the presence of strong x-ray linear dichroism as explained further in the Supplemental Material [26].
- [26] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevLett.117.147401> for experimental geometry, sample characterization, and further XAS and RIXS data and calculations, which includes Refs. [27–29].
- [27] M. Björck and G. Andersson, *J. Appl. Crystallogr.* **40**, 1174 (2007).
- [28] R. D. Cowan, *The Theory of Atomic Structure and Spectra* (University of California Press, Berkeley, 1981).
- [29] G. van der Laan, *J. Phys. Soc. Jpn.* **63**, 2393 (1994).
- [30] H.-B. Yang, J. D. Rameau, Z.-H. Pan, G. D. Gu, P. D. Johnson, H. Claus, D. G. Hinks, and T. E. Kidd, *Phys. Rev. Lett.* **107**, 047003 (2011).
- [31] L. J. P. Ament, M. van Veenendaal, T. P. Devereaux, J. P. Hill, and J. van den Brink, *Rev. Mod. Phys.* **83**, 705 (2011).
- [32] M. P. M. Dean, *J. Magn. Magn. Mater.* **376**, 3 (2015).
- [33] D. Meyers, S. Middey, M. Kareev, M. van Veenendaal, E. J. Moon, B. A. Gray, J. Liu, J. W. Freeland, and J. Chakhalian, *Phys. Rev. B* **88**, 075116 (2013).

- [34] G. Ghiringhelli, A. Piazzalunga, C. Dallera, G. Trezzi, L. Braicovich, T. Schmitt, V.N. Strocov, R. Betemps, L. Patthey, X. Wang, and M. Grioni, *Rev. Sci. Instrum.* **77**, 113108 (2006).
- [35] V.N. Strocov, T. Schmitt, U. Flechsig, T. Schmidt, A. Imhof, Q. Chen, J. Raabe, R. Betemps, D. Zimoch, J. Krempasky, X. Wang, M. Grioni, A. Piazzalunga, and L. Patthey, *J. Synchrotron Radiat.* **17**, 631 (2010).
- [36] C. H. Lai, H. S. Fung, W. B. Wu, H. Y. Huang, H. W. Fu, S. W. Lin, S. W. Huang, C. C. Chiu, D. J. Wang, L. J. Huang, T. C. Tseng, S. C. Chung, C. T. Chen, and D. J. Huang, *J. Synchrotron Radiat.* **21**, 325 (2014).
- [37] G. Catalan, *Phase Transitions* **81**, 729 (2008).
- [38] M. L. Medarde, *J. Phys. Condens. Matter* **9**, 1679 (1997).
- [39] T. Mizokawa, A. Fujimori, T. Arima, Y. Tokura, N. Mori, and J. Akimitsu, *Phys. Rev. B* **52**, 13865 (1995).
- [40] C. Piamonteze, F. M. F. de Groot, H. C. N. Tolentino, A. Y. Ramos, N. E. Massa, J. A. Alonso, and M. J. Martínez-Lope, *Phys. Rev. B* **71**, 020406 (2005).
- [41] S. Johnston, A. Mukherjee, I. Elfimov, M. Berciu, and G. A. Sawatzky, *Phys. Rev. Lett.* **112**, 106404 (2014).
- [42] J. W. Freeland, M. van Veenendaal, and J. Chakhalian, *J. Electron Spectrosc. Relat. Phenom.* **208**, 56 (2016).
- [43] V. Bisogni, S. Catalano, R. J. Green, M. Gibert, R. Scherwitzl, Y. Huang, V. N. Strocov, P. Zubko, S. Balandeh, J.-M. Triscone, G. Sawatzky, and T. Schmitt, [arXiv:1607.06758](https://arxiv.org/abs/1607.06758).
- [44] J. Jiménez-Mier, J. van Ek, D. L. Ederer, T. A. Callcott, J. J. Jia, J. Carlisle, L. Terminello, A. Asfaw, and R. C. Perera, *Phys. Rev. B* **59**, 2649 (1999).
- [45] L. D. Finkelstein, E. Z. Kurmaev, M. A. Korotin, A. Moewes, B. Schneider, S. M. Butorin, J.-H. Guo, J. Nordgren, D. Hartmann, M. Neumann, and D. L. Ederer, *Phys. Rev. B* **60**, 2212 (1999).
- [46] A. Kotani and S. Shin, *Rev. Mod. Phys.* **73**, 203 (2001).
- [47] V. I. Anisimov, D. Bukhvalov, and T. M. Rice, *Phys. Rev. B* **59**, 7901 (1999).
- [48] T. Mizokawa, D. I. Khomskii, and G. A. Sawatzky, *Phys. Rev. B* **61**, 11263 (2000).
- [49] H. Park, A. J. Millis, and C. A. Marianetti, *Phys. Rev. Lett.* **109**, 156402 (2012).
- [50] S. Johnston, A. Mukherjee, I. Elfimov, M. Berciu, and G. A. Sawatzky, *Phys. Rev. Lett.* **112**, 106404 (2014).
- [51] N. Parragh, G. Sangiovanni, P. Hansmann, S. Hummel, K. Held, and A. Toschi, *Phys. Rev. B* **88**, 195116 (2013).
- [52] E. A. Nowadnick, J. P. Ruf, H. Park, P. D. C. King, D. G. Schlom, K. M. Shen, and A. J. Millis, *Phys. Rev. B* **92**, 245109 (2015).
- [53] H. Park, A. J. Millis, and C. A. Marianetti, *Phys. Rev. B* **93**, 235109 (2016).
- [54] M. Magnuson, S. M. Butorin, A. Agui, and J. Nordgren, *J. Phys. Condens. Matter* **14**, 3669 (2002).
- [55] G. Ghiringhelli, M. Matsubara, C. Dallera, F. Fracassi, R. Gusmeroli, A. Piazzalunga, A. Tagliaferri, N. B. Brookes, A. Kotani, and L. Braicovich, *J. Phys. Condens. Matter* **17**, 5397 (2005).
- [56] G. Ghiringhelli, A. Piazzalunga, C. Dallera, T. Schmitt, V. N. Strocov, J. Schlappa, L. Patthey, X. Wang, H. Berger, and M. Grioni, *Phys. Rev. Lett.* **102**, 027401 (2009).
- [57] F. de Groot, *Coord. Chem. Rev.* **249**, 31 (2005).
- [58] F. De Groot and A. Kotani, *Core Level Spectroscopy of Solids* (CRC press, Boca Raton, 2008).
- [59] E. Stavitski and F. M. de Groot, *Micron* **41**, 687 (2010).
- [60] H. Chen, A. J. Millis, and C. A. Marianetti, *Phys. Rev. Lett.* **111**, 116403 (2013).
- [61] M. J. Han and M. van Veenendaal, *J. Phys. Condens. Matter* **26**, 145501 (2014).
- [62] P. Kuiper, J. van Elp, D. E. Rice, D. J. Buttrey, H.-J. Lin, and C. T. Chen, *Phys. Rev. B* **57**, 1552 (1998).
- [63] F. Vernay, B. Moritz, I. S. Elfimov, J. Geck, D. Hawthorn, T. P. Devereaux, and G. A. Sawatzky, *Phys. Rev. B* **77**, 104519 (2008).
- [64] L. Hozoi, L. Siurakshina, P. Fulde, and J. van den Brink, *Sci. Rep.* **1**, 65 (2011).
- [65] L. A. Wray, W. Yang, H. Eisaki, Z. Hussain, and Y.-D. Chuang, *Phys. Rev. B* **86**, 195130 (2012).
- [66] M. W. Haverkort, G. Sangiovanni, P. Hansmann, A. Toschi, Y. Lu, and S. Macke, *Europhys. Lett.* **108**, 57004 (2014).
- [67] J. Fernández-Rodríguez, B. Toby, and M. van Veenendaal, *J. Electron Spectrosc. Relat. Phenom.* **202**, 81 (2015).
- [68] P. Kuiper, D. Rice, D. Buttrey, H.-J. Lin, and C. Chen, *Physica (Amsterdam)* **208–209B**, 271 (1995).
- [69] E. Pellegrin, J. Zaanen, H.-J. Lin, G. Meigs, C. T. Chen, G. H. Ho, H. Eisaki, and S. Uchida, *Phys. Rev. B* **53**, 10667 (1996).
- [70] J. Suntivich, W. T. Hong, Y.-L. Lee, J. M. Rondinelli, W. Yang, J. B. Goodenough, B. Dabrowski, J. W. Freeland, and Y. Shao-Horn, *J. Phys. Chem. C* **118**, 1856 (2014).
- [71] C. T. Chen, L. H. Tjeng, J. Kwo, H. L. Kao, P. Rudolf, F. Sette, and R. M. Fleming, *Phys. Rev. Lett.* **68**, 2543 (1992).
- [72] E. Pellegrin, N. Nücker, J. Fink, S. L. Molodtsov, A. Gutiérrez, E. Navas, O. Strebels, Z. Hu, M. Domke, G. Kaindl, S. Uchida, Y. Nakamura, J. Markl, M. Klauda, G. Saemann-Ischenko, A. Krol, J. L. Peng, Z. Y. Li, and R. L. Greene, *Phys. Rev. B* **47**, 3354 (1993).
- [73] J. Zaanen, G. A. Sawatzky, and J. W. Allen, *Phys. Rev. Lett.* **55**, 418 (1985).
- [74] T. Mizokawa, H. Namatame, A. Fujimori, K. Akeyama, H. Kondoh, H. Kuroda, and N. Kosugi, *Phys. Rev. Lett.* **67**, 1638 (1991).
- [75] D. Khomskii, [arXiv:cond-mat/0101164](https://arxiv.org/abs/cond-mat/0101164).
- [76] M. A. Korotin, V. I. Anisimov, D. I. Khomskii, and G. A. Sawatzky, *Phys. Rev. Lett.* **80**, 4305 (1998).
- [77] T. Tsuyama, T. Matsuda, S. Chakraverty, J. Okamoto, E. Ikenaga, A. Tanaka, T. Mizokawa, H. Y. Hwang, Y. Tokura, and H. Wadati, *Phys. Rev. B* **91**, 115101 (2015).
- [78] J. Kuneš, V. Křápek, N. Parragh, G. Sangiovanni, A. Toschi, and A. V. Kozhevnikov, *Phys. Rev. Lett.* **109**, 117206 (2012).
- [79] F. C. Zhang and T. M. Rice, *Phys. Rev. B* **37**, 3759 (1988).